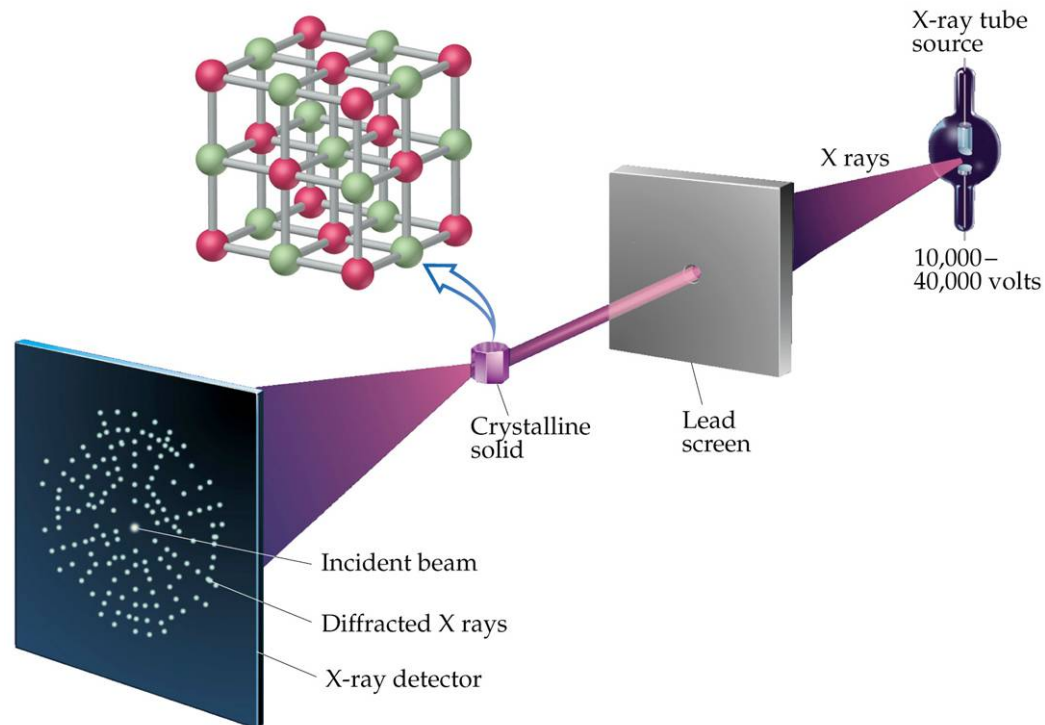
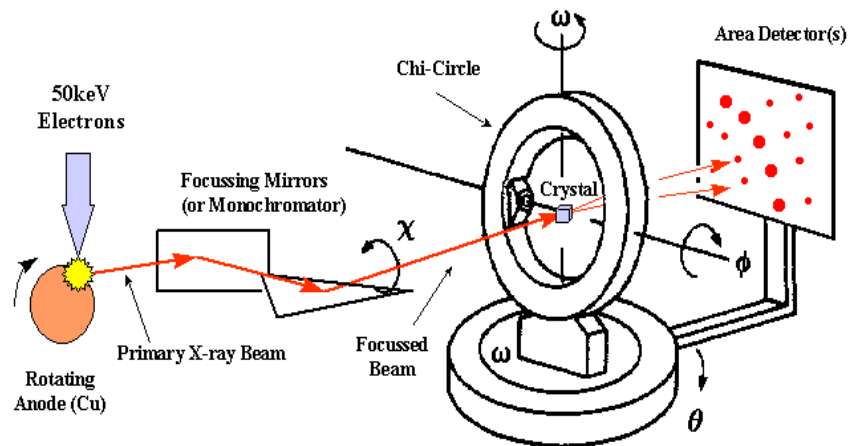


# La diffraction des rayons X

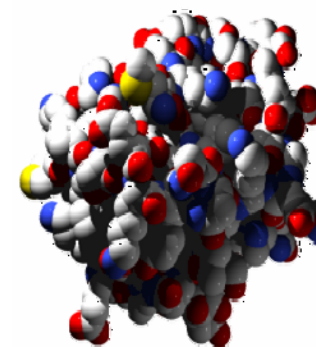
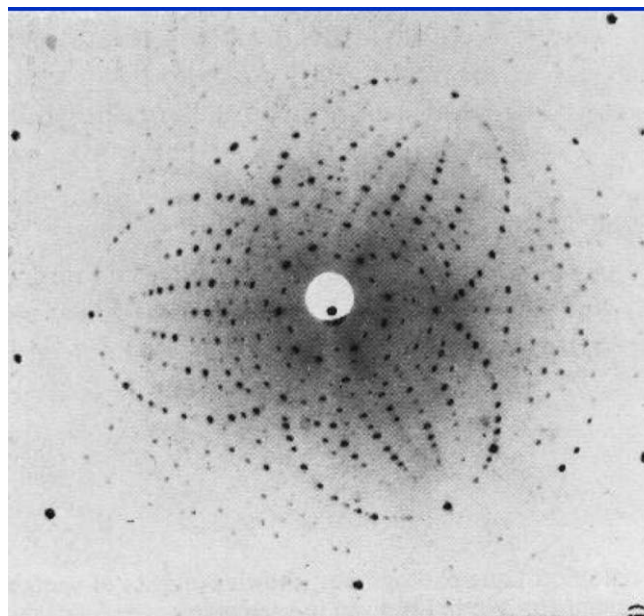
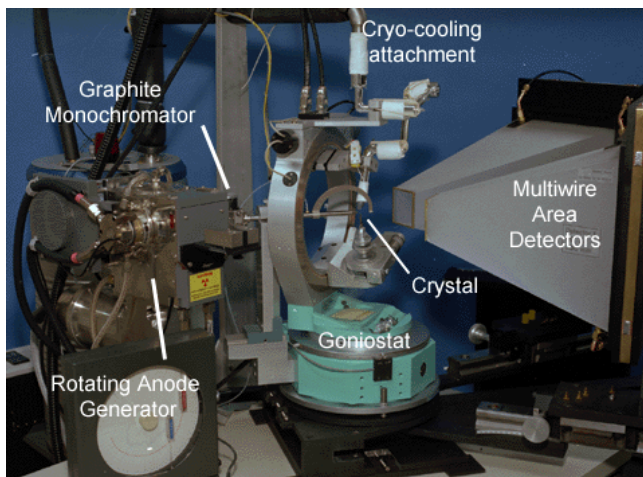
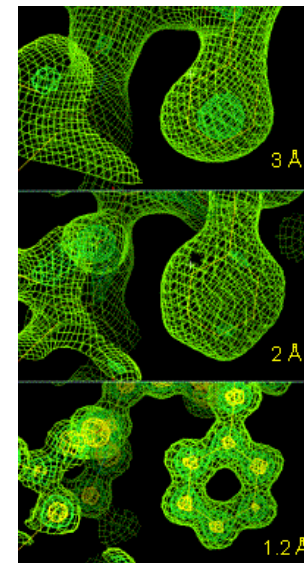
où comment retrouver la position de  $10^{23}$  atomes à partir d'un petit nombre de paramètres?



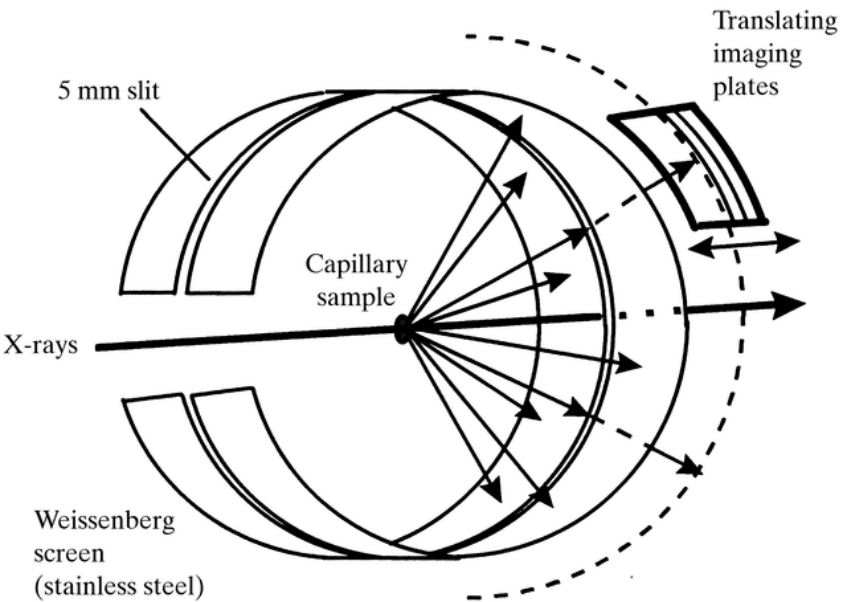
# Structure des protéines



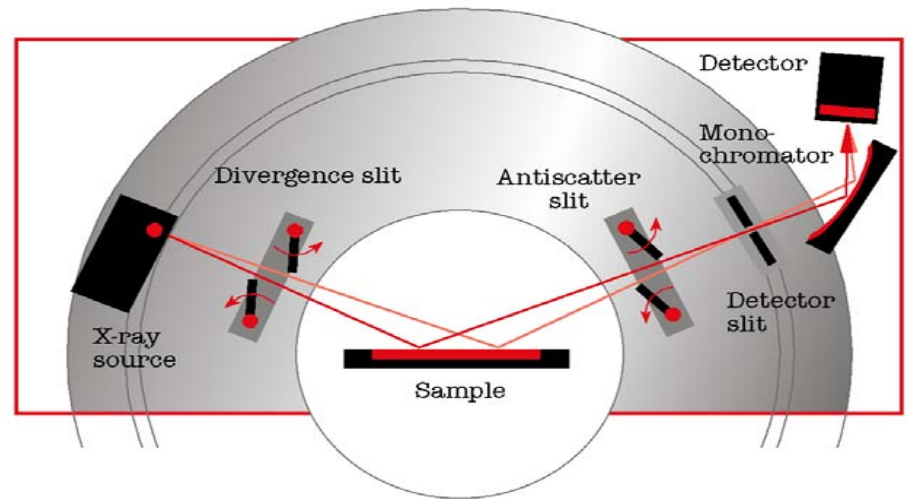
4-Circle Goniometer (Eulerian or Kappa Geometry)

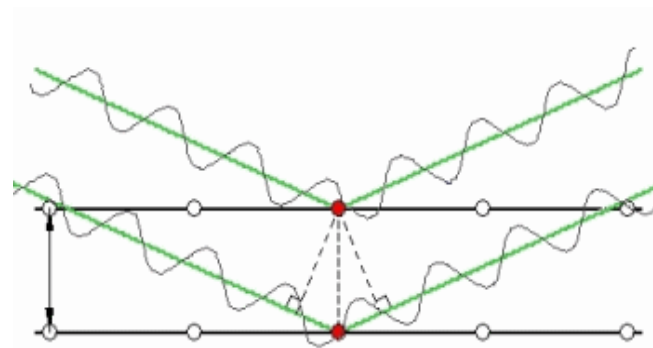
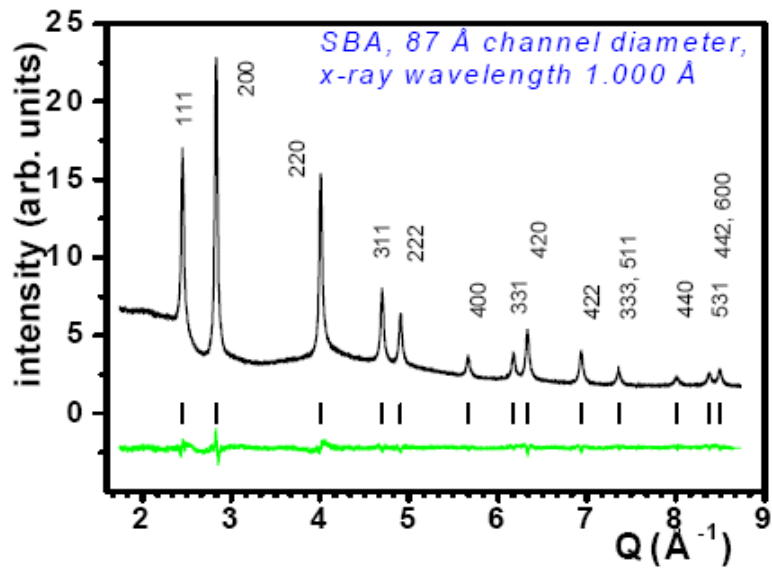


## Debye Scherrer

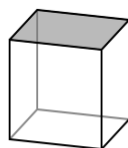


## Bragg-Brentano

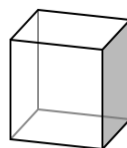




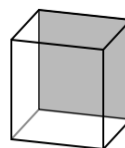
$$2d_{(hkl)}\sin\theta = n\lambda$$



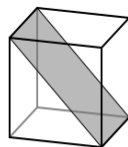
(001)



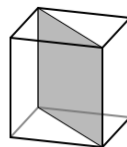
(100)



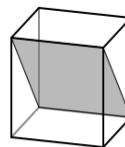
(010)



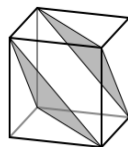
(101)



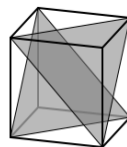
(110)



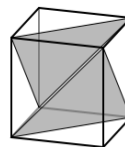
(011)



(111)



( $\bar{1}11$ )



( $\bar{1}\bar{1}1$ )

$$I_{hkl} \propto m L(\theta) P(\theta) |F_{hkl}|^2$$

m : multiplicité = nb de plans équivalents

Table 3.3. Plane Multiplicity Factors,  $M_{hkl}$

System	$hkl$	$hhl$	$hh0$	$0kk$	$hhh$	$hk0$	$h0l$	$0kl$	$h00$	$0k0$	$00l$
Cubic	$48^a$	24	12	(12)	8	$24^a$	( $24^a$ )	( $24^a$ )	6	(6)	(6)
Tetragonal	$16^a$	8	4	(8)	(8)	$8^a$	8	(8)	4	(4)	2
Hexagonal	$24^a$	$12^a$	6	(12)	(12)	$12^a$	( $12^a$ )	$12^a$	6	(6)	2
Orthorhombic	8	(8)	(8)	(8)	(8)	4	(4)	(4)	2	(2)	(2)
Monoclinic	4	(4)	(4)	(4)	(4)	(4)	(2)	(4)	2	(2)	(2)
Triclinic	2	(2)	(2)	(2)	(2)	(2)	(2)	(2)	(2)	(2)	(2)

<sup>a</sup>When all permutations of indices do not produce equivalent planes,  $M$  must be reduced by half.

P facteur de polarisation (diffusion thomson) - pour une onde non polarisée

$$P(\theta) = \frac{1 + (\cos(2\theta))^2}{2}$$

L facteur de Lorentz (dépend de la géométrie de l'expérience : intersection entre la sphere d'Ewald et le réseau réciproque)

$$L(\theta) = \frac{1}{\sin^2(\theta)\cos(\theta)}$$

$$|F_{hkl}| \text{ facteur de structure} \quad F_{hkl} = \sum_j f_j e^{2i\pi(hx_j + ky_j + lz_j)}$$

# coefficient de diffusion atomique

Where B, the Debye-Waller temperature factor is:

$$f = f_0 \exp\left[-\frac{B \sin^2 \theta}{\lambda^2}\right]$$

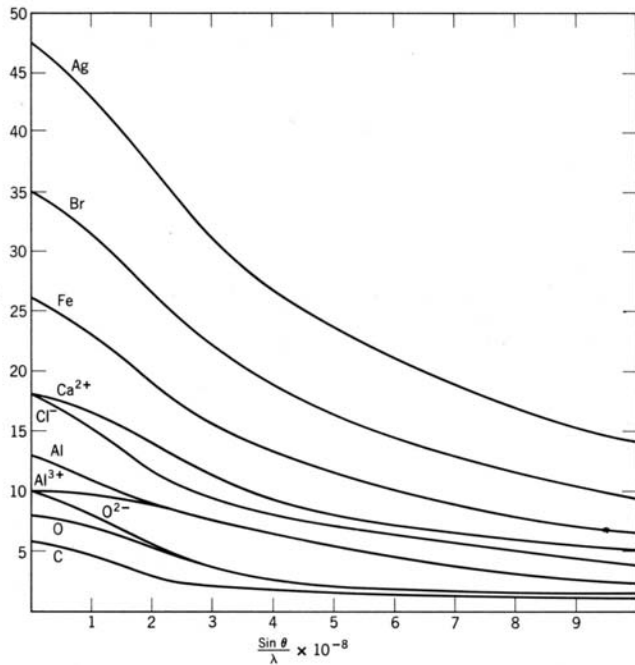


Fig. 3-15. Diffracting power of a few common atoms and ions (after C. W. Bunn, *Chemical Crystallography*, Oxford University Press, 1945).

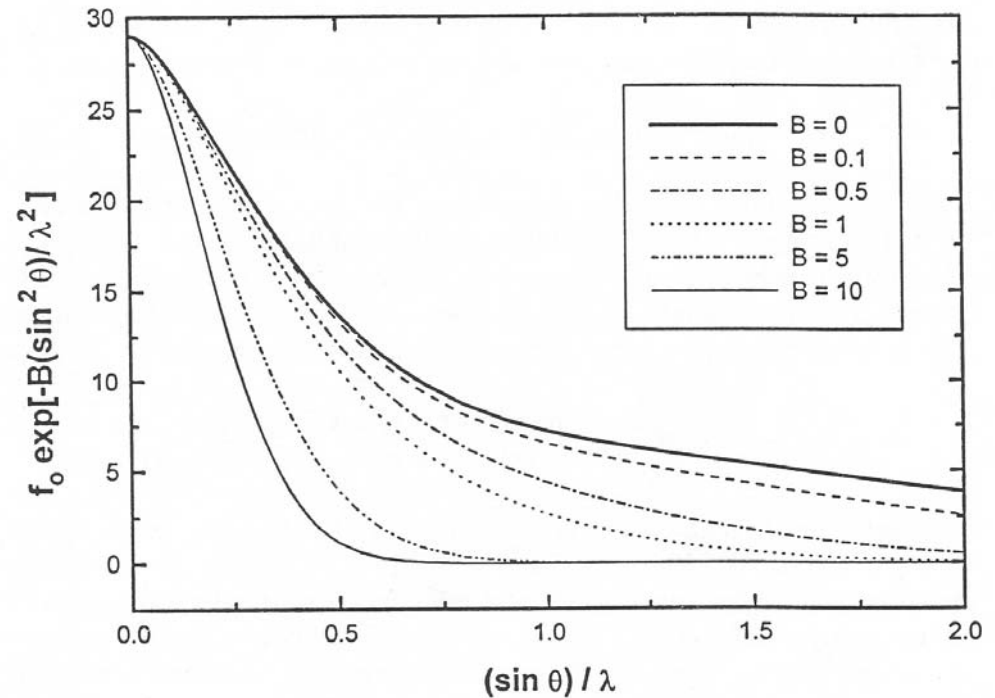
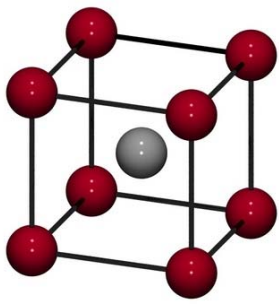


Figure 3.14. The effect of atomic thermal motion on the copper scattering factor.



## Extinctions systématiques

$$F_{hkl} = \sum_j f_i e^{2i\pi(hx_j + ky_j + lz_j)}$$

For example, for a body-centered cubic cell, for each atom located at  $x, y, z$  there will be an identical atom located at  $x+\frac{1}{2}, y+\frac{1}{2}, z+\frac{1}{2}$ . The structure factor  $F_{hkl}$  is represented by the following equation.

$$F_{hkl} = \left( \sum_{j=1}^{m/2} f_n \exp[2\pi i(hx_j + ky_j + lz_j)] \right) + \left( \sum_{n=1}^{m/2} f_n \exp\left[2\pi i\left(hx_j + \frac{h}{2} + ky_j + \frac{k}{2} + lz_j + \frac{l}{2}\right)\right] \right)$$

While complicated, it is noted that if  $h + k + l$  is even, then the second term will contain an integer,  $n$ , in it. An integral number of  $2\pi$ 's will have no effect on the value of this term and the equation reduces to:

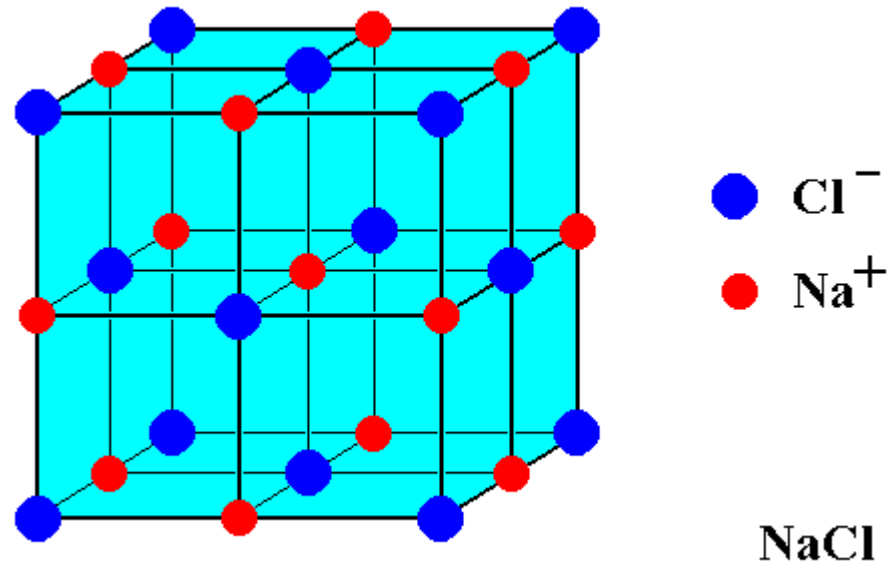
$$F_{hkl} = \sum_{j=1}^{m/2} 2f_n \exp[2\pi i(hx_j + ky_j + lz_j)]$$

If  $h + k + l$  is odd, however, the second term will contain an integer with a  $2\pi(n/2)$  term; here  $n$  is any integer and represents a full rotation of the scattering vector. This causes the second term to be negative, and the net result is there is no diffracted intensity (since  $F_{hkl} = 0$ ). This condition is called a *systematic extinction*. The table below lists the systematic extinction conditions due to translational symmetry elements<sup>1</sup>:

Symmetry	Extinction Conditions
$P$	none
$C$	$hkl; h + k = \text{odd}$
$B$	$hkl; h + l = \text{odd}$
$A$	$hkl; k + l = \text{odd}$
$I$	$hkl; h + k + l = \text{odd}$
$F$	$hkl; h, k, l \text{ mixed even and odd}$
$2_1 \parallel b$	$0k0; k = \text{odd}$

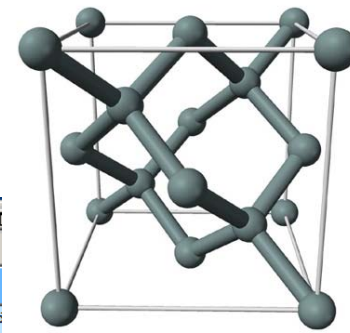
$P$  = primitive lattice;  $C, B, A$  = side-centered on  $c-, b-, a$ -face;  $I$  = body centered;  $F$  = face centered (001)

question : KCl et NaCl adoptent la même structure (NaCl). Pourtant, le diagramme de diffraction de KCl est plus proche d'un cubique simple que d'un CFC – pourquoi?





# simulation du diagramme de diffraction : poudre (silicium)



POUDRIX (15/11/03)
File Results Erase Help About Quit

Data
X rays
Neutrons
Calcul.

Source: X ray tube

**Cell parameters**

a 5.43  $\alpha$  90  
b 5.43  $\beta$  90  
c 5.43  $\gamma$  90

**Select space group**

Hermann-Mauguin: FD3M  
Hall:  
Space Group Nb.:

**Experiment**

$\lambda$  1.54056  
2 $\theta$ 1 5  
2 $\theta$ 2 110  
U 0.001  
V 0  
W 0.005

Multiple W.L.  
MoKa1  
MoKa2  
MoKmo  
MoKBet  
CuKa1  
CuKa2  
CuKmo  
CuKBet

Th.  2Th  Q

**Atom coordinates of the asymmetric unit**

Atom	Lab.	X	Y	Z	B	Pop
Si		0.12500	0.12500	0.12500	1.00	1.000

**Anomalous dispersion**

	1.5406	1.5406
Elem.	f'	f''
Si	0.000	0.000

Read Anomalous Factors

Results

Nb.	h	k	l	Dhkl	2Theta	k*Int.	lambda
K= 0.0009777							
1	1	1	1	3.1350	28.447	100.0	1.54056
2	2	0	2	1.9198	47.310	61.6	1.54056
3	1	1	3	1.6372	56.131	34.9	1.54056
4	2	2	2	1.5675	58.866	0.0	1.54056
5	0	0	4	1.3575	69.142	8.6	1.54056
6	3	1	3	1.2457	76.390	12.3	1.54056
7	4	2	2	1.1084	88.047	16.2	1.54056
8	1	1	5	1.0450	94.971	6.8	1.54056
9	3	3	3	1.0450	94.971	2.3	1.54056
10	4	0	4	0.9599	106.731	5.6	1.54056

**Diffraction diagram**

File Y Scale Print Save image Erase Difference Help

Intensity  
Calculated

20.0      40.0      60.0      80.0      100.0 2Theta

Click on a peak to index it, or click on a line of the list. Click and drag a box to "zoom". Click on the right button to "unzoom".

# JCPDS Card

Quality of data

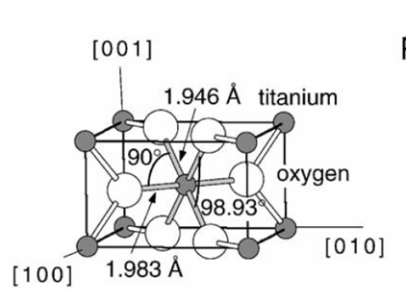
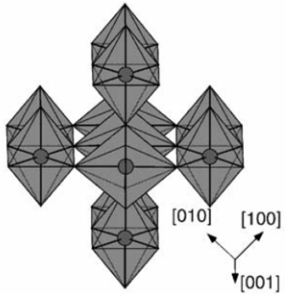
①	5-628	②	2.82	1.99	1.63	③	3.26	NaCl	④		★		
	$d$							Sodium Chloride			(Halite)		
	$I/I_1$		100	55	15		13						
⑤	Rad. CuK $\alpha_1$ $\lambda$ 1.5405 Filter Ni Dia. Cut off $I/I_1$ Diffractometer $I/I$ cor. Ref. Swanson and Fuyat, NBS Circular 539, Vol. 2, 41 (1953)							$dA$	$I/I_1$	hkl	$dA$	$I/I_1$	hkl
	Sys. Cubic S.G. Fm3m (225)							3.258	13	111			
⑥	$a_0$ 5.6402 $b_0$	$c_0$	A	C			2.821	100	200				
	$\alpha$	$\beta$	Z 4	Dx 2.164			1.994	55	220				
	Ref. Ibid.							1.701	2	311			
								1.628	15	222			
								1.410	6	400			
								1.294	1	331			
								1.261	11	420			
								1.1515	7	422			
⑦	$e\alpha$	$n\omega\beta$ 1.542	$e\gamma$	Si gn			1.0855	1	511				
	$2\nu$	D	mp	Color	Colorless			0.9969	2	440			
	Ref. Ibid							0.9533	1	531			
								0.9401	3	600			
								0.8917	4	620			
⑧	An ACS reagent grade sample recrystallized twice from hydrochloric acid.							0.8601	1	533			
	X-ray pattern at 26°C							0.8503	3	622			
	Merck Index, 8th Ed., p. 956							0.8141	2	444			
	Halite-galena-periclase group.												
												⑨	

1.file number 2.three strongest lines 3.lowest-angle line 4.chemical formula and name 5.data on diffraction method used 6.crystallographic data 7.optical and other data 8.data on specimen 9.data on diffraction pattern.

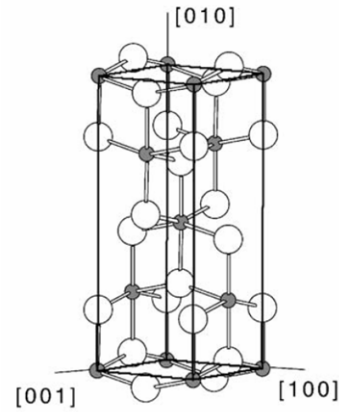
*Joint Committee on Powder Diffraction Standards, JCPDS (1969)  
Replaced by International Centre for Diffraction Data, ICDF (1978)*

# Oxyde de titane

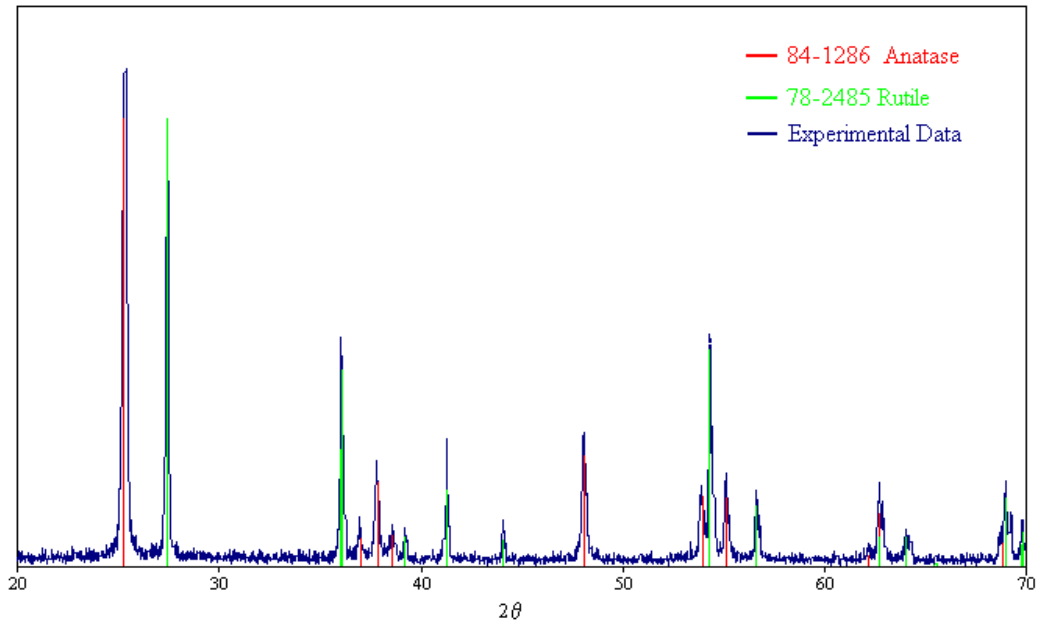
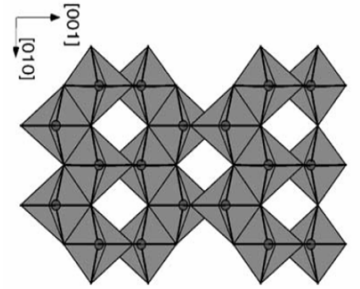
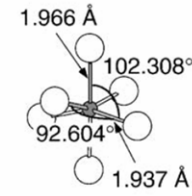
Rutile



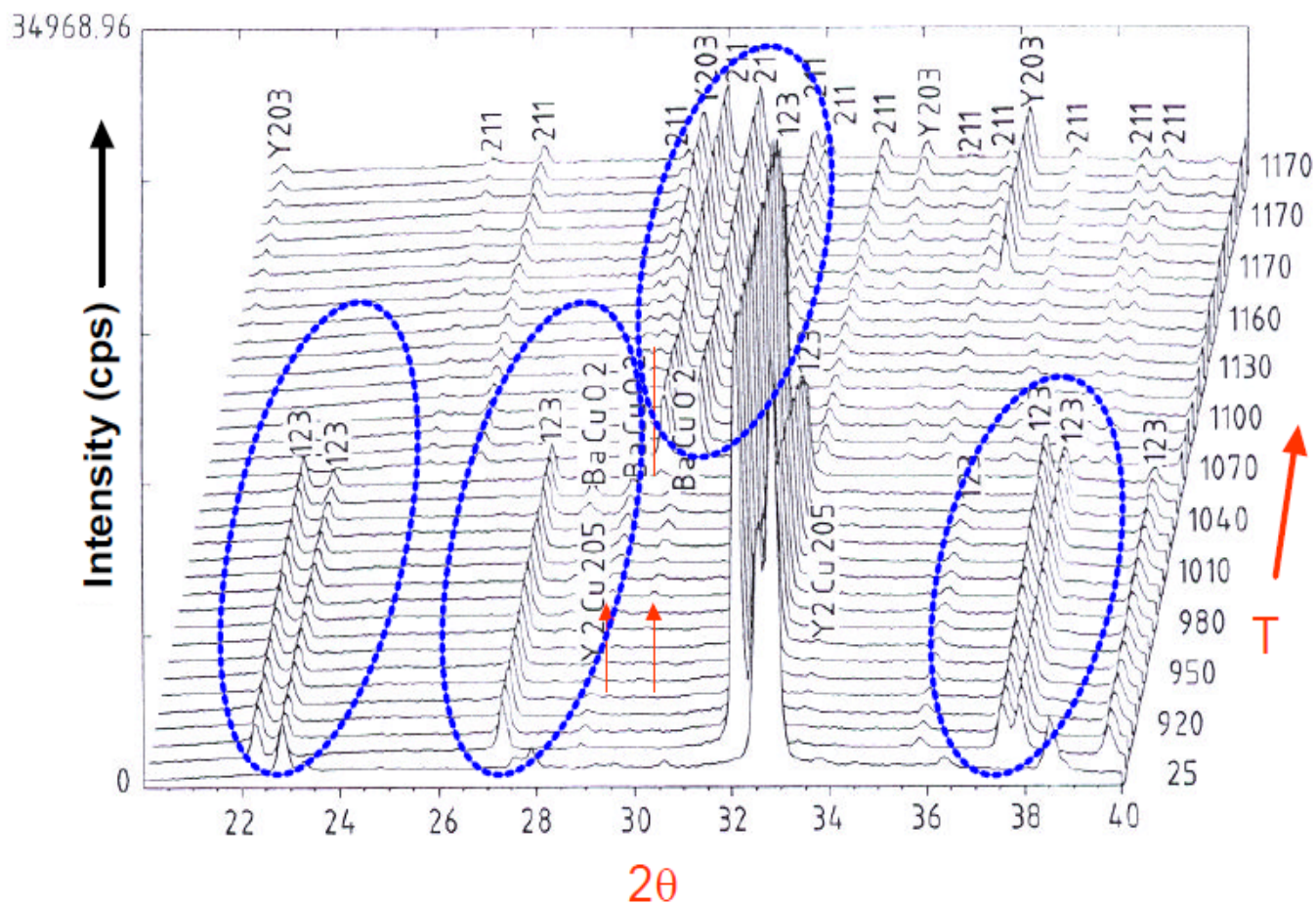
|



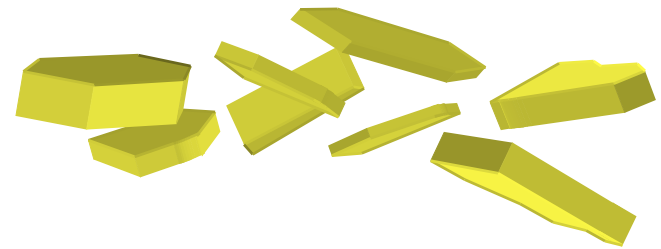
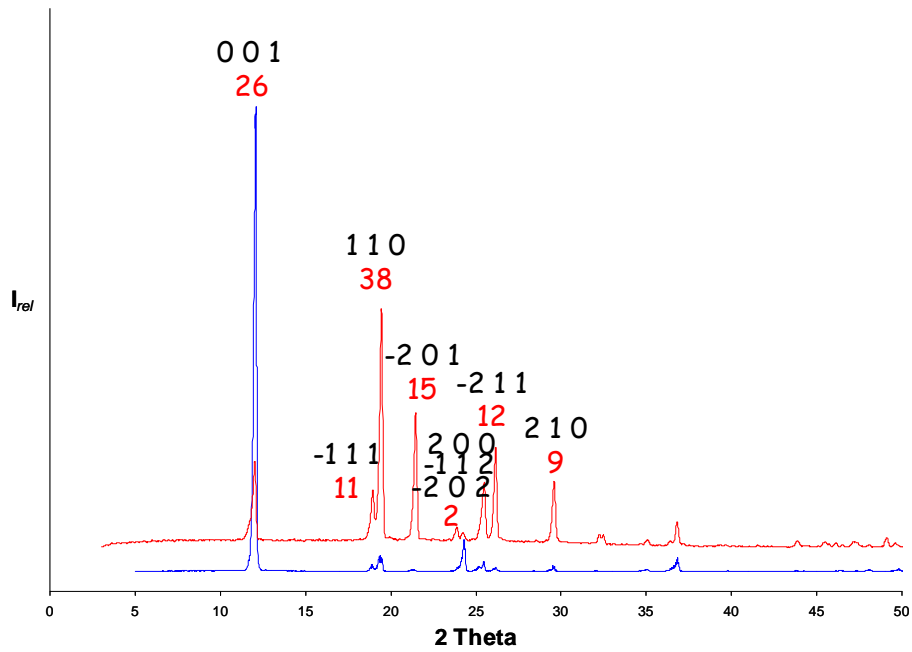
Anatase



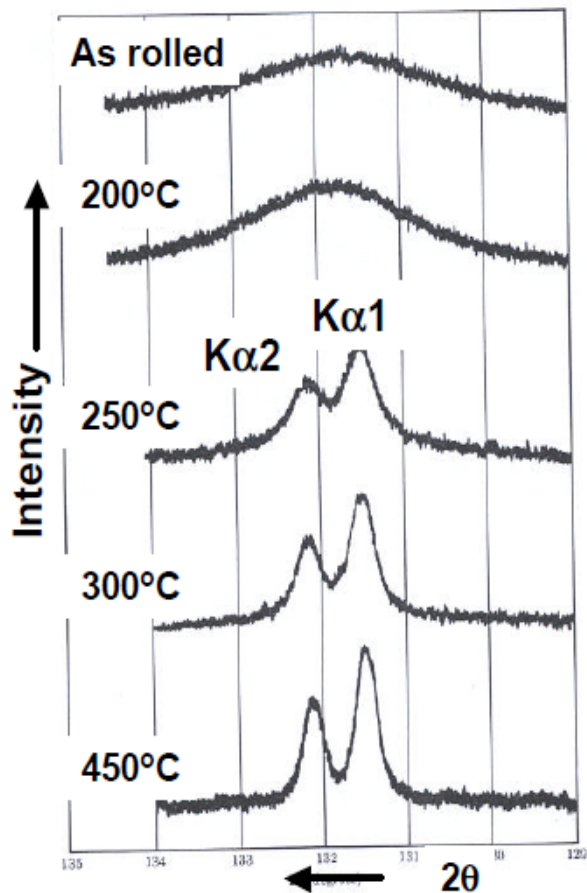
# High Temperature XRD Patterns of the Decomposition of $\text{YBa}_2\text{Cu}_3\text{O}_{7-\delta}$



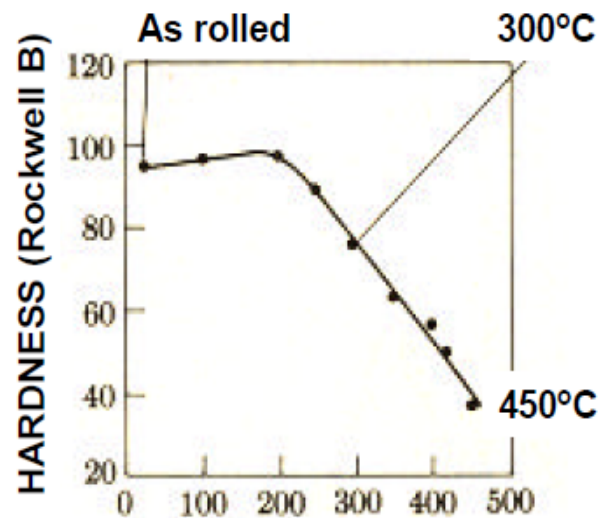
# Effet de texturation



# Effect of Coherent Domain Size



(331) Peak of cold-rolled and Annealed 70Cu-30Zn (brass)

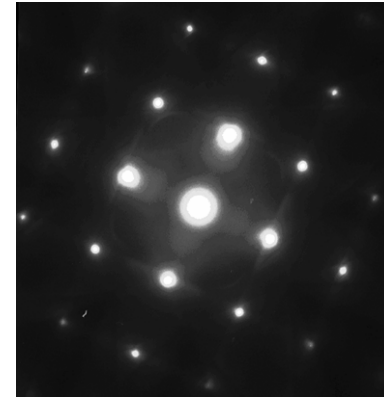


ANNEALING TEMPERATURE (°C)

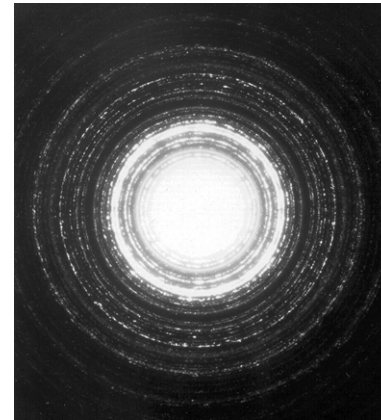
$$B = \frac{0.9 \cdot \lambda}{t \cdot \cos \theta} \quad \text{Peak Broadening Scherrer Model}$$

As grain size decreases hardness increases and peaks become broader

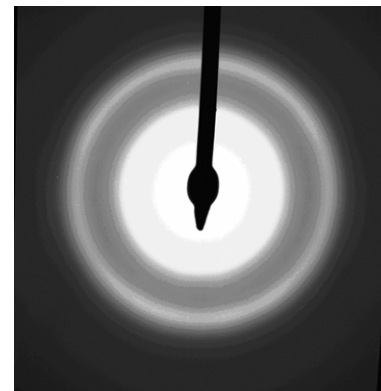
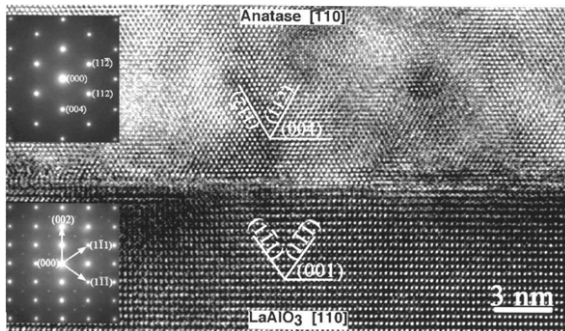
# Diffraction électronique



monocrystal

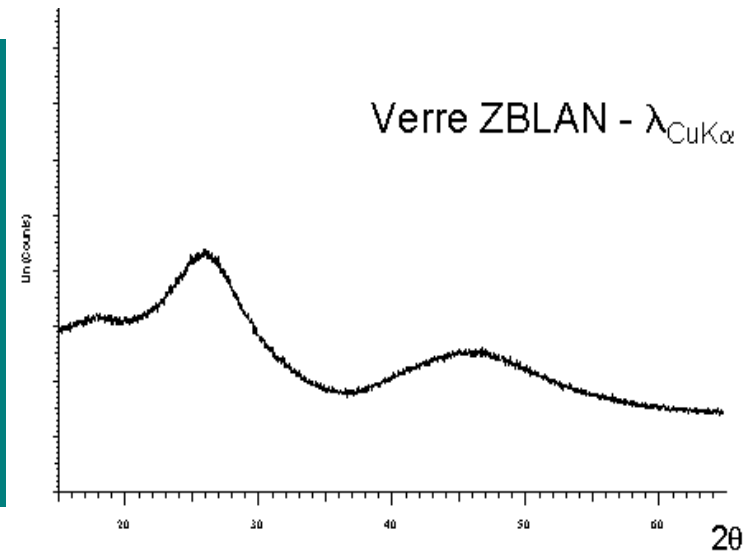
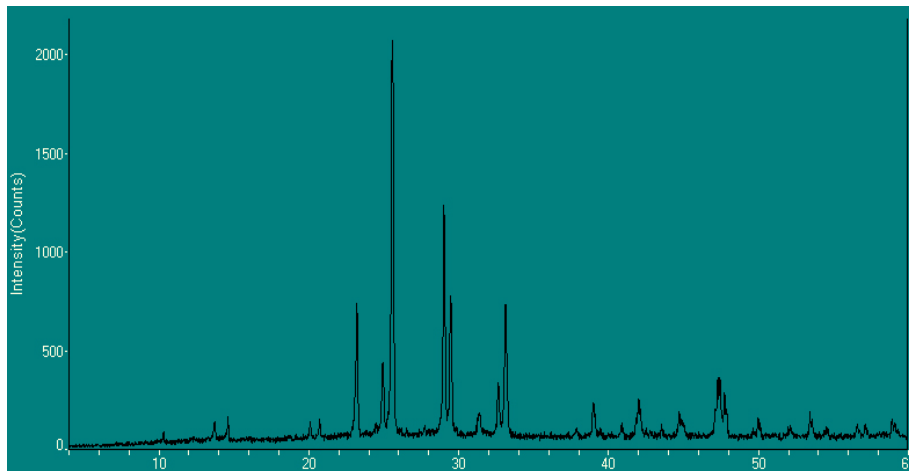
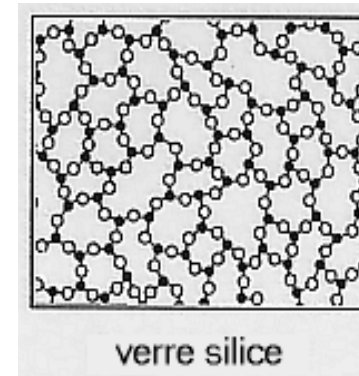
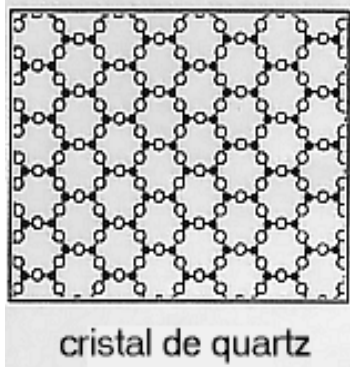


poudre



amorphe

# diffraction des amorphes





# fonction de distribution radiale

